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# Synthesis of La<sub>1−x</sub>Fe<sub>x</sub>VO<sub>4</sub> solid solutions and their photocatalytic activity

Jingang Zhao<sup>a,b</sup>, Junfeng Ma<sup>a,\*</sup>, Changhong Dai<sup>b</sup>, Ye Liu<sup>a,b</sup>, Zuwei Song<sup>a</sup>, Yong Sun<sup>a</sup>, Jingrui Fang<sup>a</sup>, Chang Gao<sup>a</sup>, Zhensen Liu<sup>a</sup>, Xia Sun<sup>b</sup>

<sup>a</sup> State Key Lab. of Green Building Materials, China Building Materials Academy, Beijing 100024, People's Republic of China <sup>b</sup> Institute of Environment and Municipal Engineering, Qingdao Technological University, Qingdao 266033, People's Republic of China

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## **1. Introduction**

Rare earth compounds of stoichiometric  $RBO<sub>4</sub>$ , where R is a rare earth and B is transition element, are well known for their wide range of physical/chemical properties. Among these materials, lanthanide orthovanadate (LaVO<sub>4</sub>) is an important rare earth compound which has been applied as laser host materials [\[1\],](#page-2-0) catalysts [\[2\], a](#page-2-0)nd luminescent materials [\[3\]](#page-2-0) due to its unique electronic structure and the numerous transition modes involving the 4f shell of rare earth ions  $[4-6]$ . Generally, LaVO<sub>4</sub> crystallizes in two polymorphs, namely, monoclinic monazite and tetragonal zircon type. As reported previously, the monoclinic-phase  $LaVO<sub>4</sub>$ is neither a suitable host for luminescent activators [\[7,8\],](#page-2-0) nor a promising catalyst [\[2\]](#page-2-0) owing to its monazite structure compared to other orthovanadates. Nevertheless, the tetragonal-phase LaVO4 possesses potential excellent properties, as revealed by some researchers [\[9\]](#page-2-0) though it is difficult to be prepared by conventional methods. A variety of methods have been explored for preparing LaVO<sub>4</sub> or LaVO<sub>4</sub> solid solutions: the traditional solid-state reaction [\[10\], s](#page-2-0)ol–gel method [\[11\], a](#page-2-0)nd hydrothermal process [\[3,12,13\],](#page-2-0) etc. Moreover, the controlled phase-transition could be realized in the hydrothermal process [\[13\]](#page-2-0) by tuning pH value of the growth solution. However, there have still existed in above approaches such drawbacks as either needing high calcining temperature

# **ABSTRACT**

Nanocrystalline La<sub>1−x</sub>Fe<sub>x</sub>VO<sub>4</sub> (x = 0, 0.05, 0.10, 0.20) solid solution powders were successfully synthesized via a low-temperature molten salt method. The products were characterized using powder X-ray diffraction (XRD), transmission electron microscopy (TEM). The results show a gradual phase transformation from monoclinic to tetragonal-phase with the increase of  $x$  value, accompanied by a remarkable morphology variation from nearly spherical nanoparticles to nanorods, nanosheets.  $Fe^{3+}$  doped LaVO<sub>4</sub> can be fully stabilized in tetragonal-phase at  $x = 0.20$ , whereas only a monoclinic-phase can be obtained without adding Fe<sup>3+</sup>. The followed photocatalytic degradation data present that the metastable tetragonal-phase exhibits higher photocatalytic activity than the monoclinic.

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and long holding time, or using complicated manipulation like repeatedly crushing-calcining for the complete reaction between reactants, or consuming a large amount of organic agents, and requiring specific equipment. Although a series of orthovanadates with nominal compositions  $La_{1-x}Fe_xVO_4$  can be synthesized by the traditional solid-state reaction [\[14\],](#page-2-0) their products, in practice, are either monoclinic LaVO<sub>4</sub> (triclinic FeVO<sub>4</sub>) or a mixture of both of them. No tetragonal-phase can be obtained. Here, we report on alternative method for preparing La<sub>1−x</sub>Fe<sub>x</sub>VO<sub>4</sub> solid solutions, under a molten salt medium condition, both the monoclinic-phase and the tetragonal can be successfully prepared at a synthetic temperature of not exceeding 450 ◦C. It fully exhibits unique advantages of the molten salt method such as simple facilities, easy manipulation, and environmentally friendly process [\[15–17\]. T](#page-2-0)he photocatalytic degradation data of the as-prepared samples also confirm the superior photocatalytic performance of the tetragonalphase La<sub>1−x</sub>Fe<sub>x</sub>VO<sub>4</sub> solid solution over that of the monoclinic LaVO<sub>4</sub>.

### **2. Experimental**

# 2.1. Synthesis of La<sub>1−x</sub>Fe<sub>x</sub>VO<sub>4</sub> Solid Solutions

Well-crystallized La<sub>1-x</sub>Fe<sub>x</sub>VO<sub>4</sub> solid solutions (x=0, 0.05, 0.10, 0.20) were prepared via a low-temperature molten salt method, the whole process can be described as follows: analytical grade NaVO<sub>3</sub>.2H<sub>2</sub>O, La(NO<sub>3</sub>)<sub>3</sub>.6H<sub>2</sub>O, and  $Fe(NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O$  were used as starting materials. Appropriate amounts of metal nitrates and  $\text{NaVO}_3 \cdot 2\text{H}_2\text{O}$  were separately weighed according to the nominal composition of La<sub>1−x</sub>Fe<sub>x</sub>VO<sub>4</sub> (x = 0, 0.05, 0.10, 0.20), and then dissolved in distilled water to form aqueous solutions. Fox each x, 3 aqueous solutions respectively contained La<sup>3+</sup>, Fe<sup>3+</sup>, and VO<sub>3</sub><sup>-</sup> ions were mixed together with strongly magnetic stirring at room temperature, and the pH value of the mixed solution was adjusted to 10.5 by

<sup>∗</sup> Corresponding author. Tel.: +86 10 51167477; fax: +86 10 65761714. E-mail address: [majunfengcbma@yahoo.com.cn](mailto:majunfengcbma@yahoo.com.cn) (J. Ma).

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<span id="page-1-0"></span>adding ammonia water; a precipitate could be obtained. The precipitate was filtered, washed with distilled water for several times, and then dried in an oven at 60 ◦C for 3 h to obtain La<sub>1−x</sub>Fe<sub>x</sub>VO<sub>4</sub> precursor. The precursor was well mixed by milling in absolute ethanol with LiNO<sub>3</sub> whose melting point is 253 °C [\[18\], w](#page-2-0)here the weight ratio of the salt to the precursor were selected as 8:1. The obtained mixture was put into an alumina crucible and calcined at 450 ◦C for 6 h. The product was dissolved in hot distilled water, and then filtered and thoroughly washed with distilled water for several times, and finally dried at 60 ◦C for 3 h.

## 2.2. Characterization

X-ray diffraction measurement was performed at room temperature by using an X-ray powder diffractometer (XRD, D8ADVANCE, Bruker, Germany) with Cu  $K_{\alpha}$ radiation ( $\lambda$  = 1.5406 Å). A transmission electron microscope (TEM, JEM-1200 EX, JEOL Ltd., Japan) was used to observe the particle size and morphology of the asprepared powders.

#### 2.3. Photocatalytic degradation evaluation

The photocatalytic activity of the typical samples was evaluated by comparing the degradation rates of Rhodamine B in aqueous solution under UV-light irradiation. 0.50 g of sample was added into a Pyrex glass beaker containing Rhodamine B aqueous solution (20 mg/L, 500 ml). Before UV-light irradiation, the suspension was stirred for 1 h under atmosphere by an air pump in a dark box. Then, it was irradiated using a 400W high pressure Hg lamp (with a major emission at about 365 nm) and continuously stirred. The reaction temperature was kept at  $25 \pm 2$  °C by a cooling water system. The periodic interval was set 10 min. 5 ml was taken from the suspension every time, and centrifuged to obtain filtrates, which were finally analyzed by a spectrophotometer (722-E, Shanghai spectrum Instrument Co. Ltd., China) at its maximum absorption wavelength of 554 nm.



**Fig. 1.** XRD patterns of  $La_{1-x}Fe_xVO_4$  samples with different values of x: (a) 0; (b) 0.05; (c) 0.10; and (d) 0.20. (Calcining temperature:  $450^{\circ}$ C; holding time: 6 h; and the weight ratio of the salt to the precursor: 8:1).

# **3. Results and discussion**

# 3.1. X-Ray diffraction analysis

Fig. 1 shows XRD patterns of La<sub>1−x</sub>Fe<sub>x</sub>VO<sub>4</sub> (x=0, 0.05, 0.10, 0.20) samples obtained by the molten salt method at 450 °C. Obvi-



**Fig. 2.** TEM images of La<sub>1−x</sub>Fe<sub>x</sub>VO<sub>4</sub> samples with different values of x: (a) 0; (b) 0.05; (c) 0.10; and (d) 0.20.

<span id="page-2-0"></span>

**Fig. 3.** Photocatalytic degradation rates of typical La<sub>1−x</sub>Fe<sub>x</sub>VO<sub>4</sub> samples (x=0, 0.20) and Rhodamine B blank solution: ( $\bullet$ ) La $_{0.80}$ Fe $_{0.20}$ VO $_4$ ; ( $\blacksquare$ ) LaVO $_4$ ; and ( $\blacktriangle$ ) Rhodamine B solution.

ously, at  $x = 0$  ([Fig. 1\(a](#page-1-0))), the XRD pattern of the as-prepared sample matches well with the reported data of monoclinic-phase  $LaVO<sub>4</sub>$ (JCPDS: 50-0367). Likewise, [Fig. 1\(d](#page-1-0)) presents a XRD pattern which can be ascribed to tetragonal-phase LaVO<sub>4</sub> (JCPDS: 32-0504) at  $x = 0.20$ . It means, without adding Fe<sup>3+</sup>, only monoclinic-phase LaVO<sub>4</sub> can be obtained; whereas at  $x = 0.20$ , Fe<sup>3+</sup> can fully stabilize LaVO<sub>4</sub> in metastable tetragonal-phase, which, in practice, is a Fe<sup>3+</sup> doping solid solution ( $La<sub>0.80</sub>Fe<sub>0.20</sub>VO<sub>4</sub>$ ). [Fig. 1\(b](#page-1-0)) and (c) (x=0.05, 0.10) is situated in a transition state from monoclinic-phase to the tetragonal. Both of them are a mixture of the monoclinicphase and tetragonal one. Moreover, with the increase of x value, the monoclinic-phase in quantity gradually declines, at  $x = 0.20$ , it completely disappears. Compared with our results, the traditional solid-state reaction route shows its obvious limitation: e.g. tetragonal-phase La<sub>1−x</sub>Fe<sub>x</sub>VO<sub>4</sub>[14] or La<sub>1−x</sub>Ce<sub>x</sub>VO<sub>4</sub> [10] solid solution cannot be formed in the whole  $x$  range.

## 3.2. TEM observation

[Fig. 2](#page-1-0) presents TEM images of above  $La_{1-x}Fe_xVO_4$  samples. The monoclinic-phase LaVO<sub>4</sub> ( $x=0$ ) nanoparticles have homogeneous and nearly spherical morphology, and their particle size is ca. 50 nm ([Fig. 2\(a](#page-1-0))). With x value increasing, the mixtures  $(x = 0.05, 0.10)$  of monoclinic-phase and tetragonal-phase are nanorods of 10–30 nm width and 30–200 nm length with perfectly crystallized morphol-ogy [\(Fig. 2\(](#page-1-0)b) and (c)). At  $x=0.20$ , well-crystallized nanosheets with ca. 50 nm width and 100–200 nm length can be formed, as shown in [Fig. 2\(d](#page-1-0)). The preferred orientation growth should be attributed to the phase transformation of  $LaVO<sub>4</sub>$  matrix solid solutions (La<sub>1−x</sub>Fe<sub>x</sub>VO<sub>4</sub>) from m-phase to t-phase. The similar results can also be found in the reference [13], where the selected-control hydrothermal synthesis of  $LAVO<sub>4</sub>$  was reported. Though m-phase LaVO<sub>4</sub> particles have nearly spherical morphology ( $pH = 2.5$ ), further increasing pH value leads to the formation of t-phase  $LaVO<sub>4</sub>$ , e.g. at pH = 4.5, well-crystallized nanorods with the preferred orientation growth can be successfully obtained.

The results suggest that  $Fe^{3+}$  doping in LaVO<sub>4</sub> matrix would promote the preferred orientation growth of  $La_{1-x}Fe_xVO_4$  solid solutions besides stabilizing the tetragonal-phase.

#### 3.3. Photocatalytic activity analysis

Fig. 3 gives the photocatalytic degradation data of Rhodamine B aqueous solution under UV-light irradiation by two typical samples ( $x = 0$ , and  $x = 0.20$ ); simultaneously, the degradation rate of Rhodamine B blank solution (without adding any catalyst) is also plotted in Fig. 3 for comparison. The monoclinic-phase  $LaVO<sub>4</sub>$ nanoparticles show a poor photocatalytic activity. Within 60 min, about 52% photodegradation rate can be achieved, i.e. only a little higher than that of the blank solution. However,  $La<sub>0.80</sub>Fe<sub>0.20</sub>VO<sub>4</sub>$ solid solution with the pure tetragonal-phase exhibits much higher photocatalytic activity than the monoclinic LaVO<sub>4</sub>; within the same time of 60 min, its photogradation rate reaches over 95%. The superior photocatalytic activity of the tetragonal-phase over the monoclinic-phase can also be found in the other references [19], which was further confirmed in present study.

## **4. Conclusion**

Homogeneous and well-crystallized nanocrystalline La<sub>1−x</sub>Fe<sub>x</sub>VO<sub>4</sub> solid solutions can be synthesized at 450 °C by a low-temperature molten salt method. Their phase composition can be easily tuned by varying x value. At  $x = 0$ , only a monoclinicphase can be obtained, whereas pure tetragonal-phase can be prepared even at  $x = 0.20$ . The mixtures of both monoclinic-phase and tetragonal-phase appear in a range of  $0 < x < 0.20$ . Fe<sup>3+</sup> doping in LaVO<sub>4</sub> matrix can not only stabilize La<sub>1-x</sub>Fe<sub>x</sub>VO<sub>4</sub> solid solution in metastable tetragonal-phase at a relatively lowtemperature (450 $\degree$ C) but also favor their preferred orientation growth. The tetragonal-phase solid solution exhibits much higher photocatalytic activity than the monoclinic.

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